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# Thiocyanate-Modified Silver Nanofoam for Efficient CO<sub>2</sub> Reduction to CO

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 $cm^{-2}$  and the mass activity of 23.5 A  $g_{Ag}^{-1}$ , which are the highest values among recently reported Ag-based electrocatalysts. Mechanism studies reveal that the catalytic performance of the AgNF correlates with the density of surface SCN ligands, which exhibit excellent electrochemical stability under negative potentials. Density functional theory calculations suggest that SCN ligands promote the formation of COOH\* intermediates by modifying the local electron density at the active sites. Further, the synthesis method is applicable to different catalyst substrates. For example, the AgNF grown on a carbon-based gas diffusion film exhibits an ultrahigh mass activity of 52.1 A  $g_{Ag}^{-1}$  and maintains its high CO selectivity simultaneously, demonstrating excellent potentials for practical applications.

**KEYWORDS:** silver, nanofoam, thiocyanate, CO<sub>2</sub> reduction, electrodeposition

# INTRODUCTION

The fast increasing atmospheric carbon dioxide  $(CO_2)$ concentration is a critical challenge for human society.<sup>1,2</sup> A sustainable solution is to electrochemically reduce CO<sub>2</sub> to useful chemical species, such as carbon monoxide (CO), formic acid, methanol, ethylene, or methane, using electrical energy generated from renewable sources.<sup>3-10</sup> In particular, CO can be used as fuels or feedstocks for producing chemicals via the Fischer-Tropsch process.<sup>11,12</sup> The electrochemical reduction of CO<sub>2</sub> to CO follows a two-electron reduction pathway:  $CO_2 + 2H^+ + 2e^- = CO + H_2O$  with a relatively low redox potential of -0.53 V versus the standard hydrogen electrode (SHE). Hydrogen evolution reaction (HER) in aqueous electrolytes is a critical competitive reaction of CO<sub>2</sub> reduction reaction (CO<sub>2</sub>RR), which significantly compromises the selectivity to CO and reduces the efficiency of CO2RR devices.<sup>13</sup> An ideal electrocatalyst for CO<sub>2</sub>RR is expected to deliver a large current density and maintain a high selectivity toward desired reduction products.

Silver (Ag) is a promising electrocatalyst for  $CO_2RR$  to CO due to its relatively low cost and high activity. A simple Ag foil can achieve a Faradaic efficiency toward CO (FE<sub>CO</sub>) of more than 60%.<sup>14</sup> Recent studies show that the performance of Ag-

based electrocatalysts can be significantly improved by tailoring their physical structures and surface chemical compositions.<sup>14–21</sup> For physical structures, the catalytic activity of Ag nanoparticles (AgNPs) displays a volcano-shaped trend for CO<sub>2</sub>RR over the size of the AgNPs.<sup>19</sup> Triangular Ag nanoplates with Ag(100) facets show superior selectivity and activity as compared to AgNPs.<sup>22</sup> Highly curved Ag nanochannels created by an alloying–dealloying method have a high FE<sub>CO</sub> of 90% at -0.8 V<sub>RHE</sub>.<sup>16,23</sup> At the optimal thickness of ~6  $\mu$ m, nanoporous Ag thin films can deliver a maximum Ag mass activity of 12.2 A g<sub>Ag</sub><sup>-1.24</sup> For surface chemical composition modifications, Ag surfaces modified by Cl<sup>-</sup>, S<sup>2-</sup>, or phosphate anions<sup>17,21,25</sup> or small molecules, such as cysteamine,<sup>26</sup> ethylenediamine, oleylamine, or 1-dodecanethiol,<sup>27</sup> have demonstrated improved selectivity to CO than unmodified Ag surfaces. The stability of surface modification agents on Ag surfaces under cathodic potentials is essential in

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maintaining the high CO selectivity.<sup>28</sup> Current surface modification agents can only retain the high CO selectivity (>90%) at relatively low potentials, that is,  $-0.9 V_{RHE}$ , affording the highest mass activity of 2.56 A  $g_{Ag}^{-1}$  from cysteamine-modified AgNPs.<sup>26</sup> We noticed that thiocyanate (SCN) exhibits strong interactions with Ag, and it is electrochemically stable even under negative potentials.<sup>29</sup> Further, as a pseudohalide, SCN may introduce covalency to electrocatalysts, which could be beneficial for the adsorption of intermediates in CO<sub>2</sub>RR.<sup>28–31</sup> However, SCN has not been used to modify Ag electrocatalysts for CO<sub>2</sub>RR.

Herein, we first used a dynamic hydrogen bubble-templated electrodeposition method to create a free-standing threedimensional porous Ag nanofoam (denoted as AgNF),<sup>32,33</sup> which is engraved with abundant curved surfaces and surface SCN ligands. The catalytic performance of this AgNF for electrochemical reduction of  $CO_2$  was evaluated together with several reference Ag electrocatalysts, including Ag nanoparticles (AgNPs), Ag foils, and AgNFs with different densities of surface SCN ligands. The roles of SCN ligands were investigated by Raman spectroscopy. Further, density functional theory (DFT) calculations were carried out to gain a better understanding of the catalytic function of AgNF.

### EXPERIMENTAL SECTION

Electrodeposition of SCN-Modified AgNF. Ag foil (99.99%, 0.1 mm thick, Sigma-Aldrich) was first cleaned by washing with diluted HNO3. Next, SCN-modified AgNF was electrodeposited on the cleaned Ag foil in a three-electrode configuration on an electrochemical workstation (CHI 660E). A Pt gauze and a Ag/AgCl electrode (3 M KCl) acted as the counter and reference electrodes, respectively. The plating area is  $0.5 \times 0.5$  cm<sup>2</sup> on one side of the Ag foil while an insulating adhesive sealed the other side. A mixture of  $Ag_2SO_4$  (99.99%, Sigma, 0.01 M), KSCN (99%, Sigma, 1.5 M), and (NH<sub>4</sub>)<sub>2</sub>SO<sub>4</sub>, (99.9%, Sigma, 0.5 M) aqueous solutions served as the electrolyte. The deposition current density was -1 A cm<sup>-2</sup>, and the deposition time was 20 s. It should be noted that a two-chamber cell was used for electrodeposition of AgNF. The cells are physically separated by glass frits (5 mm) to avoid possible Pt contamination. Pt is absence from AgNF as confirmed by X-ray photoelectron spectroscopy (XPS) spectra in the Pt 4f region. After the electrodeposition, the AgNF was further cycled in the electrolyte between -0.5 and 0.2 V versus Ag/AgCl at a scan rate of 50 mV  $s^{-1}$  for 30 cycles before it was rinsed with deionized (DI) water and dried in a vacuum overnight. A silver mass loading of  $\sim 1.45$  mg cm<sup>-2</sup> was determined by weighing the electrode before and after deposition. The electrodeposition of AgNF on the gas diffusion electrode was performed under the same condition.

**Synthesis of AgNPs.** AgNPs were synthesized using a previously reported method.<sup>17</sup> In brief, the AgNO<sub>3</sub> solution (100 mL of 10 mg mL<sup>-1</sup>) was mixed with citrate trisodium solution (100 mL of 5 mg L<sup>-1</sup>) under stirring. After 30 min, NaBH<sub>4</sub> solution (50 mL of 20 mg mL<sup>-1</sup>) was added. The resulting solid materials were recovered after 2 h, washed with DI water, and then tip sonicated in 250 mL of DI water for 2 h before the Ag NP suspension was further centrifuged at 4000 g for 1 h. The resulting supernatant was drop cast on a carbon cloth to reach a mass loading of 1.45 mg cm<sup>-2</sup>, matching the mass loading of AgNF on the Ag foil electrodes.

Material Structural Characterization. Physical structures of synthesized materials were examined by a scanning electron microscope (SEM, Joel JSM-6700F) and a transmission electron microscope (TEM, FEI Themis-Z). Their Raman spectra were collected on a Raman microscope (Renishaw inVia) using a 785 nm laser. X-ray diffraction (XRD) patterns of AgNF deposits scratched off from substrates were recorded on an XRD diffractometer (Bruker AXS D8). XPS spectra were obtained on a Thermo Scientific K-Alpha spectrometer with a Al K $\alpha$  (1486.3 eV) source.

**Fabrication of AgNF with Reduced SCN Surface Density.** The electrochemical reduction of AgNF was carried out in an Ar-saturated KHCO<sub>3</sub> electrolyte (0.5 M) using the same three-electrode configuration used for electrodeposition. The reduction was performed under two different potentials for 30 min before the AgNF electrode was rinsed with DI water and dried in ambient condition. The electrodes reduced at -1.0 and -1.4 V<sub>RHE</sub> were denoted as AgNF-R1 and AgNF-R2, respectively.

Electrochemical reduction of CO<sub>2</sub> and Product Analysis. All CO<sub>2</sub> electrochemical reduction tests were carried out on the electrochemical workstation (CHI 660E) using a gas-tight H-shaped two-compartment cell separated by a preactivated membrane (Sigma-Aldrich, Nafion 117). Each compartment contains 60 mL of 0.5 M KHCO3 as the electrolyte, and the headspace is about 30 mL. The electrolyte was stirred at 400 rpm using a PTFE-covered magnetic bar. The electrolyte was first saturated with Ar and pre-electrolyzed with a Pt electrode at  $-0.3 V_{RHE}$  for 30 min before saturation with CO<sub>2</sub> for the tests. A Pt gauze and a Ag/AgCl (3 M KCl) electrode served as the counter and reference electrodes, respectively, in one compartment, while a AgNF electrode was used as the working electrode in the other compartment. Chronoamperometric current-time (i-t) curves were recorded at different potentials. All potentials were calibrated to RHE by adding  $0.210 + 0.0591 \times pH$ . The pH value of the electrolyte after saturation with CO<sub>2</sub> was measured by a pH meter (Horiba, PC1100). Tafel plots were compiled from stable CO specific current densities at various potentials after 2 h testing. The gaseous products were analyzed on a gas chromatograph (Agilent 6890 N) using PLOT MolSieve 5A and Q-bond PLOT capillary column with He (99.999%) as the carrier gas.

Determination of Electrochemical Active Surface Area (ECSA). The ECSA of Ag electrodes was determined by comparing electrons required to form a monolayer of Ag<sub>2</sub>O or Ag(OH) on the Ag surface in a three-electrode configuration using Ar-saturated 0.1 M KOH as the electrolyte.<sup>16</sup> A Ag/AgCl (3 M KCl) electrode and a Pt gauze were used as the reference and counter electrodes, respectively. The working Ag electrodes were first electrochemically reduced at -0.8 V<sub>RHE</sub> for 10 min and rinsed with DI water. Afterward, they were oxidized at 1.15 V<sub>RHE</sub> in a fresh 0.1 M KOH electrolyte for 10 min, during which a monolayer of Ag was oxidized to form Ag<sub>2</sub>O or Ag(OH). The charge required to form such a monolayer is about 400  $\mu$ C cm<sup>-2</sup>. The current– time (*i*–*t*) curves for the oxidation of AgNF, AgNP, and Ag foil electrodes were recorded.

**Density Functional Theory (DFT) Calculations.** All DFT calculations were conducted using the Vienna ab initio simulation package (VASP). The electronic exchange was described by a generalized gradient approximation method. The electronic correlation was described with a Perdew–Burke–Ernzerhof functional.<sup>34</sup> Core electrons were described by a projector augmented-wave method.<sup>35</sup> The valance

electrons were described by expanding the Kohn-Sham wave functions in a plane wave basis.<sup>35</sup> Gamma point sampling of the Brillouin zone was used for the calculations of Ag clusters. The Brillouin zone for the model of Ag(111) was sampled using a  $(3 \times 3 \times 1)$  Monkhorst-Pack k-point mesh and integrated using the method by Methfessel and Paxton.<sup>36</sup> The energy cutoff for the calculations was set at 400 eV, and the force convergence criteria were set as 0.05 eV Å<sup>-1</sup>. Spin polarization was tested and was found to have a negligible influence on the results. Zero-point energy and entropic corrections (with a temperature of 298 K) were applied to the calculated reaction energies. The integrated heat capacity corrections were applied using the values obtained by Gao and co-workers.37 The initial geometry of the Ag clusters was assumed to have a 147-atom cuboctahedral shape. Thiocyanates were modeled at the truncated corner sites, similar to a previous study.<sup>26</sup> The Ag(111) surface was modeled as a threelayer  $(5 \times 5)$  slab with the bottom layer fixed in bulk position.

## RESULTS AND DISCUSSION

A dynamic  $H_2$  bubble-templated electrodeposition method was used to synthesize AgNF with a hierarchical porous structure.<sup>38</sup> The electrodeposition was carried out under a current density of  $-1 \text{ A cm}^{-2}$  for 20 s on one side of prepolished Ag foils. The electrodeposition of Ag was accompanied by vigorous  $H_2$ bubble formation, coalescence, and dissipation, resulting in AgNF with interconnected mesopores and macropores, which avoids the use of solid templates and their removals.  $(NH_4)_2SO_4$  was used as the electrolyte for electrodeposition to prevent the possible incorporation of halide ions (e.g.,  $Cl^-$ ).<sup>32,33</sup> After the electrodeposition, the shining Ag foils were coated by a layer of AgNF in grey color with a mass loading of 1.45 mg cm<sup>-2</sup>. AgNPs with a diameter of ~10 nm were also prepared as a reference for comparison.Figure S1 in the Supporting Information shows their SEM and TEM images.

The morphology of AgNF was first examined by SEM. Figure 1a reveals abundant macropores with a diameter of about 20  $\mu$ m, which were created by the dissipation of large H<sub>2</sub>



Figure 1. Morphology and chemical composition of AgNF. (a, b) SEM images of AgNF at different magnifications. (c, d) Bright-field TEM images at different magnifications. The inset of (c) shows the corresponding diffraction pattern, and the inset of (d) is the Fourier-transformed image. (e) HAADF-STEM image and its corresponding EDX elemental mapping results. Ag and S signals are overlaid for comparison.

bubbles. These macropores not only expose the inner space of the AgNF but also can facilitate mass transportation of molecules/ions in electrolytes and gas bubbles during CO<sub>2</sub>RR. The higher magnification SEM image in Figure 1b displays interconnected Ag ligaments, which form mesopores of less than 50 nm in diameter. These pores with highly curved surfaces are similar to those reported previously using an alloying-dealloying method, which demonstrated high selectivity in reducing CO<sub>2</sub> to CO.<sup>16</sup> We further characterized the structure of the AgNF by TEM. The inset of Figure 1c displays the selected area diffraction pattern of Ag ligaments with various diffraction rings, which can be assigned to multiple Ag facets. Figure 1d displays TEM images collected at a higher magnification. The Ag lattice fringes with a *d*-spacing of 0.23 nm are from Ag(111). The corresponding Fourier-transformed image shows a highly symmetric pattern, suggesting an extended crystalline network. A thin layer (~1 nm) with poor crystallinity was found at the Ag ligament surface, which exhibited reduced contrast under the high-angle annular darkfield mode scanning TEM (HAADF-STEM, Figure 1e) as a result of surface modification. The corresponding elemental mappings obtained by the energy-dispersive X-ray spectroscopy (EDX) confirm the existence of S, C, N, and O on the surface Ag ligaments. S and N distribute uniformly on the Ag surface. The overlaid Ag and S signals indicate that S is enriched at the surface. Overall, the TEM characterization results indicate the existence of surface-anchored S, C, and N species.

The XRD pattern of AgNF was obtained by measuring Ag deposits scratched from Ag foils to avoid strong XRD signals from the substrates. Figure 2a shows four predominant diffraction peaks  $(2\theta = 38.2^\circ, 44.4^\circ, 64.5^\circ, \text{ and } 77.4^\circ)$ , which originate from various Ag crystal facets (PDF#04-0783), agreeing with the SAED pattern shown in Figure 1c. The inset of Figure 2a shows the magnified XRD pattern at the low diffraction angle region  $(2\theta = 10^{\circ} \text{ to } 40^{\circ})$ . The weak peak of Ag<sub>2</sub>O ( $2\theta = 32.8^\circ$ , PDF#41-1104) suggests some surface oxidation by O<sub>2</sub> dissolved in electrolytes or oxidation in air. No XRD features belonging to bulk AgSCN (PDF#36-0608) or AgCl (PDF#22-1326) are observed. Raman and X-ray photoelectron spectroscopies were used to elucidate the chemical configurations of S, C, and N. Figure 2b displays the Raman spectrum of AgNF, showing three features at  $\delta_{SCN}$ (443 cm<sup>-1</sup>),  $v_{CS}$  (723 cm<sup>-1</sup>), and  $v_{CN}$  (2112 cm<sup>-1</sup>) originated from thiocyanate.<sup>39</sup> Some weak Raman features at 1000–1500  $cm^{-1}$  can be attributed to surface Ag<sub>2</sub>O. Meanwhile, there are no Raman features from Ag–S compounds, such as Ag<sub>2</sub>S,  $Ag_2SO_3$ , or  $Ag_2SO_4$ .<sup>40,41</sup> The XPS survey scan spectrum in Figure 2c indicates the existence of S, C, and N on the AgNF surface with the abundance of 2.19, 1.96, and 2.28 at %, respectively, which indicate an average coverage of 0.1884 SCN per surface Ag atom (see Figure S2 and the related calculation in the Supporting Information). The highresolution XPS spectra of S, C, and N in Figure 2d are similar to those of the reference AgSCN (Figure S3 in the Supporting Information). Besides, no features from S-O are observed in the high binding energy region ( $\sim 168 \text{ eV}$ ), indicating that S atoms exist in SCN ligands on the AgNF. Moreover, the featureless XPS spectrum collected in the Cl 2p region rules out the presence of Cl in the AgNF (Figure S4 in the Supporting Information).

The performance results of AgNF, AgNPs (1.45 mg cm<sup>-2</sup>), and Ag foil for CO<sub>2</sub>RR were compared in a 0.5 M KHCO<sub>3</sub>



Figure 2. (a) XRD pattern and (b) Raman spectrum of AgNF. (c) XPS spectrum of a survey scan and (d) the high-resolution XPS spectra of C, N, and S.



**Figure 3.** Performance of three types of Ag electrodes, i.e., AgNF, AgNP, and Ag foil, for electrochemical CO<sub>2</sub>RR. (a) Total current density ( $j_{tob}$ ) top) and  $j_{CO}$  (bottom), and (b) FE<sub>CO</sub> obtained at different potentials. (c) Tafel plots constructed from the measured CO partial current densities. (d) Stability tests of AgNF performed at -0.9, -1.1, and -1.3 V<sub>RHE</sub>.

electrolyte by measuring their choronopotentiometric discharges under different potentials (-0.3 to -1.5 V<sub>RHE</sub>) over a 2 h period. Gaseous products were analyzed by gas chromatography every 30 min. CO and H<sub>2</sub> were identified as the main reaction products. Small amounts of CH<sub>4</sub>, C<sub>2</sub>H<sub>4</sub>, and

formate were also detected when the applied potential is more negative than -1.4 V<sub>RHE</sub>. Figure 3a shows the quasipolarization curves, which were constructed using steady-state currents extracted from i-t curves collected under different applied potentials. The AgNF outperforms both



**Figure 4.** Characterization and performance of CO<sub>2</sub> electrochemical reduction of AgNF-R1 and AgNF-R2. (a) Raman spectra of AgNF collected after CO<sub>2</sub> electrochemical reduction under different potentials. (b) XPS survey scan spectra of AgNF-R1 and AgNF-R2. (c)  $FE_{CO}$  and (d)  $j_{CO}$  of AgNF, AgNF-R1, and AgNF-R2 under different potentials.

AgNP and Ag foil in the potential window from -0.2 to -1.5 $V_{RHE}$ , demonstrating much higher total density  $(j_{tol})$  and CO specific current density  $(j_{CO})$ . Typical current-time (i-t)curves collected at -1.2 V were shown in Figure S5 in the Supporting Information together with their calculated  $FE_{CO}$ . We also compared the  $FE_{CO}$  and  $FE_{H2}$  of AgNF, AgNPs, and Ag foil in Figure S6 in the Supporting Information. The AgNF delivered a high  $j_{tol}$  of  $-33.4 \text{ mA cm}^{-2}$  together with a high  $FE_{CO}$  of 94.8  $\pm$  2.9%. Both values are much higher than those of AgNPs (-18.4 mA cm<sup>-2</sup> and 39.0  $\pm$  3.3 %) and Ag foil (-6.1 mA cm<sup>-2</sup> and 12.6  $\pm$  3.9%). FE<sub>CO</sub> at different potentials is plotted in Figure 3b. CO can be detected in gaseous products at merely  $-0.3 V_{\text{RHE}}$  on the AgNF, which is 0.1 and 0.3 V less than those of AgNP and Ag foil, respectively. The  $FE_{CO}$  of AgNF increased quickly to 90% at -0.5  $V_{RHE}$  and maintained around 95% when the potential went down to -1.2V<sub>RHE</sub>. In comparison, both AgNPs and Ag foil displayed not only lower maximum FE<sub>CO</sub> but also narrower potential windows in which they can maintain the high CO selectivity. The AgNF exhibits the highest  $j_{CO}$  of  $-33.0 \text{ mA cm}^{-2}$  at -1.3V<sub>RHE</sub>, and it can work in a significantly extended the operation window and deliver the high current density while maintaining high selectivity to CO. Further, the corresponding mass activity of 23.5 A  $g_{Ag}^{-1}$  toward CO is one of the highest among recently reported Ag-based electrocatalysts for CO<sub>2</sub>RR.<sup>16,17,20</sup> We further compared  $j_{CO-ECSA}$  of Ag electrodes by normalizing their CO specific current density obtained at  $-0.6 V_{RHE}$  to their surface roughness factor (Figure S7 in Supporting Information).<sup>16</sup> The  $j_{\text{CO-ECSA}}$  of AgNF is -33.6  $\mu$ A cm<sup>-2</sup>, which is 9 and 60 times higher than those of AgNPs ( $-3.7 \mu$ A  $cm^{-2}$ ) and Ag foil (-0.54  $\mu$ A  $cm^{-2}$ ), respectively. This result indicates that AgNF has significantly improved intrinsic catalytic activity (see Table S2 in the Supporting Information for comparison with recently reported Ag-based electrocatalysts).

The CO<sub>2</sub> reduction kinetics on the Ag electrodes was analyzed using their Tafel plots. Figure 3c shows that AgNF has a Tafel slope of 62 mV dec<sup>-1</sup>, which is much lower than those of AgNPs at 92 mV dec<sup>-1</sup> and Ag foil at 128 mV dec<sup>-1</sup>, indicating better reduction kinetics. The commonly proposed reaction mechanism of electrochemical reduction from CO<sub>2</sub> to CO involves the formation of reaction intermediates, such as CO<sub>2</sub>\*<sup>-</sup> and COOH\*, following the four steps shown below

$$CO_2 + e^- + * \rightarrow CO_2^{*-}$$
 (1)

$$\mathrm{CO}_2^{*-} + \mathrm{H}^+ \to \mathrm{COOH}^*$$
 (2)

$$COOH^* + H^+ + e^- \rightarrow CO^* + H_2O \tag{3}$$

$$CO^* \rightarrow CO + *$$
 (4)

The Tafel slope of 62 mV dec<sup>-1</sup> from AgNF suggests the rate-limiting step is the proton-coupled step (eq 3). While the larger Tafel slopes of AgNPs and Ag foil indicate that the rate-limiting step is the initial surface adsorption of  $\text{CO}_2$  (eq 1).<sup>17,23</sup> The catalytic stability of AgNF was also evaluated by measuring their choronopotentiometric discharges under -0.9, -1.1, and -1.3 V<sub>RHE</sub> over 10 h. Figure 3d shows that the current density decreases merely 2.3, 4.5, and 9.6%, respectively. Further, FE<sub>CO</sub> remains mostly unchanged throughout the tests under -0.9 and -1.1 V<sub>RHE</sub>. However, after the 8 h test under -1.3 V<sub>RHE</sub>, FE<sub>CO</sub> drops to 77.1%. SEM images of AgNF after the stability tests are shown in Figure S8 in the Supporting Information. There are minor structural



Figure 5. Reaction mechanism investigated by DFT calculations. (a) Ag catalyst models used in the DFT calculations. (b) Charge density redistribution on the Ag surfaces anchored by SCN ligands. (c) Free energy pathways of CO<sub>2</sub> reduction at the equilibrium potential (-0.11 V) on Ag<sub>135</sub>(SCN)<sub>12</sub>, Ag<sub>143</sub>(SCN)<sub>4</sub>, Ag<sub>147</sub>, and Ag(111) surfaces. (d) The difference in limiting potentials ( $\Delta U_L$ ) of CO<sub>2</sub> reduction and HER on the Ag catalyst surfaces.

changes, except for the slight aggregation of Ag ligaments after the test under  $-1.3 V_{\text{RHE}}$ .

We further studied the potential roles of SCN ligands in improving the intrinsic catalytic activity of AgNF. The stability of SCN ligands anchored on the Ag surface at different potentials was first evaluated by Raman spectroscopy. As shown in Figure 4a, the intensity of the Raman peak of SCN ligands (i.e., the  $\nu_{\rm CN}$  peak at 2112 cm<sup>-1</sup>) has a minor decrement when the applied potential drops to  $-1 V_{\rm RHE}$ , indicating the excellent stability of SCN ligands. Further decreasing the potential leads to the fast intensity drop of the  $\nu_{\rm CN}$  peak.

The Raman peak diminishes under -1.4 V<sub>RHE</sub>, which coincides with the quickly deteriorated FE<sub>CO</sub> shown in Figure 3b. To examine the relationship between SCN ligands anchored on the Ag surface and the catalytic activity of AgNF for CO<sub>2</sub> reduction, two additional AgNF electrodes with reduced SCN ligand densities were prepared by cathodically reducing AgNF electrodes at -1.0 and -1.4 V<sub>RHE</sub> in Arsaturated 0.5 M KHCO3 electrolytes, denoted as AgNF-R1 and AgNF-R2, respectively. The XPS survey scan spectra of AgNF-R1 and AgNF-R2 in Figure 4b confirm their surface composition changes. The S concentration on AgNF-R1 drops to 0.41 at %, which is about 19% of the S concentration on AgNF before the cathodic reduction. The C and N concentrations on AgNF-R1 also decreased. In comparison, the XPS signals of S, C, and N are undetectable on AgNF-R2, suggesting that all SCN ligands were removed. The highresolution Ag 3d XPS spectra of AgNF electrodes in Figure S9 in the Supporting Information show the formation of a Ag<sup>0</sup> surface after the removal of SCN ligands and the reduction of surface oxide layers upon the reduction treatment. Figure S10 shows that high-resolution XPS spectra of S, C, and N of AgNF-R1 are similar to that of AgNF, indicating that they still exist in the thiocyanate configuration. HAADF-STEM images in Figure S11 show that the morphologies of Ag ligaments in

AgNF and AgNF-R1 are similar. The EDX mapping results also agree with the XPS results, displaying reduced S concentrations. Besides, since the SCN ligands exhibit a high electrochemical stability in a wide range of cathodic potentials, we propose that the SCN ligands would only desorb from the Ag surface without substanial structure transformation.<sup>42</sup> This is supported by the fact that no signature Raman features from other Ag–S compounds, such as Ag<sub>2</sub>S and Ag<sub>2</sub>SO<sub>4</sub>, have been observed on AgNF-R1 and AgNF-R2 (see Figure S12 in the Supporting Information).<sup>41</sup>

Next, we compared the CO<sub>2</sub> reduction performance of AgNF-R1 and AgNF-R2 at -0.7, -0.9, -1.1, and -1.3 V<sub>RHE</sub>. Their FE<sub>CO</sub> values were plotted in Figure 4c in comparison with that of AgNF. Under the mild reduction potentials of -0.7 and -0.9 V<sub>RHE</sub>, the three electrodes have similar FE<sub>CO</sub>. However, FE<sub>CO</sub> values of AgNF-R1 and AgNF-R2 drop significantly when more negative reduction potentials are applied. As listed in Table S3 in the Supporting Information, the FE<sub>CO</sub> of AgNF-R1 declines to 86.5% that of AgNF under  $-1.1 \text{ V}_{\text{RHE}}$ , and the FE<sub>CO</sub> of AgNF-R2 drops to 56.7%. When the potential decreases to  $-1.3 V_{RHE}$ , their FE<sub>CO</sub> values further drop to 56.7 and 32.4%, respectively. Figure 4d shows a similar trend in their  $j_{CO}$  values. AgNF has the highest  $j_{CO}$  of -33.0mA cm<sup>-2</sup> at -1.3 V<sub>RHE</sub>; in comparison, the highest  $j_{CO}$  values of AgNF-R1 and AgNF-R2 are only -21.3 and -16.6 mA cm<sup>-2</sup>, respectively. ECSAs of the three types of electrodes were measured to evaluate the potential influence of their different porous structures. As shown in Figure S13, the ECSAs of AgNF-R1 and AgNF-R2 drop by 2.8 and 14.3% compared with that of AgNF. The relatively small morphology changes are unlikely the main reason for the observed drastic drop in  $FE_{CO}$ and  $j_{CO}$  under the potentials of -1.1 and -1.3 V<sub>RHE</sub>. Table S4 also shows that the changing trend of normalized  $j_{CO-ECSA}$  of the three types of AgNF electrodes under different potentials is similar to that of  $j_{CO}$ . Based on the above experimental results, we conclude that anchoring SCN ligands on the Ag surface is



Figure 6. (a) Photo and (b) SEM image of the AgNF/GDE. (c) Schematic illustration of the  $CO_2RR$  electrolyzer used in this study. (d) The steady-state current density, (e) selectivity, and mass activity toward CO of the AgNF/GDE electrolyzer obtained at different potentials.

the main reason for the observed  $CO_2$  reduction catalytic activity enhancement of AgNF.

DFT calculations were carried out to gain mechanistic insights into the catalytic activity of AgNF for CO<sub>2</sub>RR. According to the famous Sabatier principle, reactant binding energies to metal catalyst surfaces are often correlated with their catalytic activity.<sup>43'-45</sup> The rate-determining step of CO<sub>2</sub> reduction to CO is the formation of COOH\* intermediates. Therefore, the adsorption free energy of COOH\* ( $\Delta G_{COOH}$ ) has been used as a reactivity descriptor to evaluate the activities of  $CO_2RR$  catalysts.<sup>22,26,46</sup> A Ag<sub>147</sub> cluster model was adopted in this study since it has shown sufficient to evaluate those much larger NPs by providing suitable undercoordinated sites for CO<sub>2</sub>RR.<sup>26,27</sup> As shown in Figure 5a, Ag particles anchored with SCN ligands were set up to represent the surfaces of AgNF and AgNF-R1 by replacing Ag atoms on corner sites of the  $Ag_{147}$  cluster with 12 (denoted as  $Ag_{135}(SCN)_{12}$ ) or 4 SCN ligands (denoted as  $Ag_{143}(SCN)_4$ ). We also assessed if it would be possible to form  $Ag(SCN)_2$  complex by linking two SCN ligands at a single Ag site. However, our calculation results obtained after relaxation suggest such a structure is unstable. Besides, it is noteworthy that the surface coverage of SCN in the cluster models  $(0.15 \text{ for } Ag_{135}(SCN)_{12} \text{ and } 0.0455 \text{ for}$  $Ag_{143}(SCN)_4$  is comparable to the experimentally determined values of 0.1884 for AgNF and 0.0358 for AgNF-R1, which shows a good agreement between theoretical models and realistic surfaces. A Ag135 cluster without SCN ligands was used to represent the AgNF-R2 sample. Considering the much weaker adsorption of the SCN on the Ag(111) surface than the undercoordinated Ag sites in the Ag clusters, it can be expected that the SCN would exhibit negligible influence on its CO<sub>2</sub>RR performance.<sup>46</sup> Therefore, a clean Ag(111) surface was adopted as a reference in DFT calculations.

Figure 5b shows that SCN ligands cause charge redistribution on Ag surfaces. Compared with  $Ag_{143}(SCN)_4$ , Ag<sub>135</sub>(SCN)<sub>12</sub> is expected to have a higher density of localized unpaired electrons, which is beneficial for stabilizing surfaceadsorbed CO<sub>2</sub><sup>-</sup> radicals and promoting their subsequent transformation to COOH\* after accepting a proton.<sup>26</sup> This calculation result correlates with the higher catalytic activity of AgNF observed in our experimental results. The calculated free energy diagram for CO<sub>2</sub> reduction toward CO on these Ag models is shown in Figure 5c. The binding energy of COOH\* ( $\Delta G_{\rm COOH}$ ) on Ag<sub>135</sub>(SCN)<sub>12</sub> is lower than those on Ag<sub>147</sub> and Ag(111), indicating improved COOH\* adsorption on Ag<sub>135</sub>(SCN)<sub>12</sub>. It should be noted that the binding energy of CO\* ( $\Delta G_{\rm CO}$ ) is similar on the Ag surfaces because CO adsorption follows a donor–acceptor mechanism instead of the radical adsorption mechanism.<sup>30</sup>

We further investigated HER on these Ag surfaces to understand the improved selectivity toward CO on AgNF. The thermodynamic limiting potentials of CO $_2$ RR and HER ( $\Delta U_{
m L}$ =  $U_{L(CO2)} - U_{L(H2)}$ ) on different surfaces were compared to explain their reaction selectivity where  $U_{\rm L}$  is the lowest potential required for all reaction steps to go downhill in the free energy diagram of  $CO_2RR$  and  $HER^{47,48}$  Our DFT results (Figure S14 in Supporting Information) suggest that although  $Ag_{135}(SCN)_{12}$  has stronger H\* binding energy, it exhibits the lowest  $\Delta U_{\rm L}$  (Figure 5d), which correlates with the experimentally observed high CO selectivity of AgNF. It is also noteworthy that the density of SCN ligands is critical. The values of  $\Delta U_{\rm L}$  gradually decrease on Ag<sub>143</sub>(SCN)<sub>4</sub> and Ag<sub>135</sub> clusters, suggesting reduced selectivity in CO<sub>2</sub>RR. Overall, our DFT calculation results agree with the experiments where the selectivity to CO in CO<sub>2</sub>RR on these Ag catalysts follows the order of AgNF > AgNF-R1 > AgNF-R2 > AgNP > Ag foil.

Other than depositing the AgNF on Ag foils, we have also grown AgNF on a carbon-based gas diffusion film electrode with a size of  $\sim 1.77$  cm<sup>2</sup> (denoted as AgNF/GDE) by the same synthesis method used for growing AgNF on Ag foils.

Figure 6a shows a photo of AgNF/GDE. AgNF/GDE with larger sizes can be prepared using a high-current power source. The Ag mass loading on the AgNF/GDE is slightly increased to 2.55 mg cm<sup>-2</sup> compared to AgNF on Ag foils because the carbon film has a rough surface. An SEM image in Figure 6b displays Ag alignments similar to those on Ag foils. Besides, XPS survey scans also show identical surface S and N abundances (i.e., 2.49 and 2.08 at %, respectively) in AgNF on the two types of substracts. These results indicate that the synthesis method of AgNF presented in this study is applicable to different catalyst substrates.

As illustrated in Figure 6c, the CO<sub>2</sub>RR performance of AgNF/GDE was assessed in a two-chamber electrolyzer separated with a Nafion 117 membrane. The AgNF/GDE side is open to a CO<sub>2</sub> atmosphere with a partial pressure of 1 atm. The steady-state current densities obtained at different potentials are displayed in Figure 6d. The gas diffusion film helps to overcome the issue related to gaseous CO<sub>2</sub> dissolution and diffusion in aqueous electrolytes, leading to a 4.1–7.5 time increment in the current density in a potential window from -0.4 to -1.3 V<sub>RHE</sub>. Moreover, Figure 6e shows that the excellent selectivity toward CO can be retained. AgNF/GDE can deliver a maximum mass activity of 52.1 A  $g_{Ag}^{-1}$  at -1.2 V<sub>RHE</sub>, demonstrating an excellent potential for efficient CO production from CO<sub>2</sub>RR.

## CONCLUSIONS

In summary, we have demonstrated a fast ( $\sim 20$  s) and efficient electrodeposition method to synthesize hierarchically porous Ag nanofoams with curved surfaces modified by a SCN ligand. The used electrolyte is an aqueous mixture of Ag<sub>2</sub>SO<sub>4</sub>, KSCN, and (NH<sub>4</sub>)<sub>2</sub>SO<sub>4</sub>. SCN ligands anchored on Ag surfaces demonstrate excellent electrochemical stability under negative potentials. Serving as an electrocatalyst for CO<sub>2</sub>RR, AgNF can maintain a high CO selectivity ( $FE_{CO} > 90\%$ ) in a wide operation window from -0.5 to -1.2 V<sub>RHE</sub> and delivers a maximum  $j_{CO}$  of 33.0 mA cm<sup>-2</sup> or a mass activity of 23.5 A  $g_{Ag}^{-1}$ , which is one of the best catalytic performance among recently reported Ag-based electrocatalysts for CO2RR. Experimental results indicate that SCN ligands are crucial in realizing the high CO selectivity in the wide operation potential window. DFT calculations suggest that the roles of SCN ligands are to promote the formation of COOH\* intermediates by inducing charge redistribution on the Ag surfaces. Moreover, the synthesis method of AgNF can be extended to other catalyst substrates. AgNF grown on carbonbased gas diffusion films can overcome the issue related to CO<sub>2</sub> dissolution and diffusion in liquid electrolytes, delivering ~2.5 times higher mass activity. This efficient and straightforward Ag catalyst has excellent potentials to be applied in practical CO<sub>2</sub> reduction devices.

### ASSOCIATED CONTENT

#### **Supporting Information**

The Supporting Information is available free of charge at https://pubs.acs.org/doi/10.1021/acscatal.9b04633.

SEM and TEM images of AgNPs; estimation of SCN coverage on AgNFs; XPS spectra of AgSCN and Cl 2p region; typical *j*-*t* curves and  $FE_{CO}$  values of AgNF, AgNP, and Ag foil obtained at -1.2 V vs RHE;  $FE_{CO}$  and  $FE_{H2}$  of Ag catalysts; ECSA measurements; properties and CO<sub>2</sub> electrochemical reduction perform-

ance at -0.6 V vs RHE of various electrodes; performance comparison of various Ag electrocatalysts for CO<sub>2</sub> reduction in aqueous electrolytes; SEM images of the AgNF after 2 h discharging at various potentials; high-resolution XPS spectra of Ag 3d in various samples; HAADF-STEM images, the corresponding EDX elemental maps, Raman spectra, ECSA, FE<sub>CO</sub>, and CO specific current densities of AgNF-R1 and AgNF-R2; and the free energy diagram of HER on different Ag models (PDF)

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### Notes

The authors declare no competing financial interest.

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